

Thermal decomposition of the Pb, Al-hydrotalcite material

M. L. VALCHEVA-TRAYKOVA, N. DAVIDOVA

Institute of Kinetics and Catalysis, Bulgarian Academy of Sciences, 11 G. Bonchev Street, Sofia 1040, Bulgaria

A. H. WEISS

Department of Chemical Engineering, Worcester Polytechnic Institute, 100 Institute Road, Worcester, MA 01609, USA

The transformation of methane into C₂ hydrocarbons over thermally degraded Mg, Pb, Al-hydrotalcite material proceeds with the active participation of β-PbO. The conditions of pre-treatment leading β-PbO formation in the catalyst have been examined on the Pb, Al-hydrotalcite material as a model system. By use of X-ray diffraction, scanning electron microscopy, transmission electron microscopy, infrared spectroscopy and diffuse-reflectance spectroscopy techniques, it was found that the temperature interval 600–750 °C is optimal for β-PbO crystallization. On the basis of the present and previous results, the formation of all components of the catalytically active thermally degraded Mg, Pb, Al-hydrotalcite material, has been explained. During calcination of the Mg, Pb, Al-hydrotalcite material, the lead-containing components form β-PbO and γ-Al₂O₃ and the magnesium-containing components transform to a finely dispersed MgO-Al₂O₃ matrix.

1. Introduction

Lead oxide participates in the electron transfer during oxidative coupling of methane over several lead oxide-containing catalysts [1]. It is assumed that Pb²⁺ is active in the formation of the CH₃ radicals on the catalytic surface [1, 2]. The 750 °C calcined Mg, Pb, Al-hydrotalcite material is reported [2] to be a good catalyst for oxidative dimerization of methane. In the X-ray diffraction (XRD) pattern of the activated catalyst, α- and β-PbO are detected in the highly dispersed matrix of MgO and γ-Al₂O₃. The β-PbO is assumed to be the most active lead oxide phase in the oxygen transfer in the partial oxidation of methane to ethane over the 750 °C calcined Mg, Pb, Al-hydrotalcite material. During the reaction, PbO transformed into lead, which on interaction with the gaseous oxygen, deoxidized to β-PbO. The deactivated catalyst contained metallic lead in a highly dispersed MgO-Al₂O₃ matrix. After regeneration, lead transformed to β-PbO and significant reflections of α-Al₂O₃ appeared in the X-ray diffraction pattern of the sample.

In the present work, the effect of the thermal treatment on the solid–solid phase transitions of the model Pb, Al-hydrotalcite material was investigated, to determine the conditions of calcination leading to β-PbO crystallization. The next purpose, on the basis of the present and previous [3] investigations, was to clarify the role of the lead and magnesium components in the formation of the catalytically active methane dimerization mixed oxide system by calcination of Mg, Pb, Al-hydrotalcite material at 750 °C. The model

system contained only lead and aluminium and was synthesized by the procedure used for the synthesis of the Mg, Pb, Al-hydrotalcite material [4].

2. Experimental procedure

2.1. Preparation of the material

The sample was prepared to contain Me²⁺:Me³⁺ atomic ratio typical for the hydrotalcite (at about 2), using the hydrotalcite preparation procedure that avoids filtering and washing problems associated with gel precipitates [4]. First, 82.75 g Pb(NO₃)₂·24H₂O (p.a.) and 47 g Al(NO₃)₃·9H₂O (p.a.) were dissolved in 350 ml de-ionized water. Separately, 35 g NaOH were dissolved with 175 ml de-ionized water in a teflon container. Then 25 g Na₂CO₃ were dissolved in de-ionized water and added to the NaOH solution in the teflon reactor. The Pb, Al solution was added to the reactor by strong mixing using a teflon stirrer, for 18 h at 32 °C. Then the temperature in the reactor was increased to 65 °C and the material was vigorously mixed at this temperature for 18 h. The vessel was removed from both mixing and heat treatment and placed in an ice bath to quench the reaction. The slurry was then suction filtered, washed and dried at 120 °C in a furnace for 24 h. The large ionic radius of Pb²⁺ determines the small probability of the formation of lead-containing hydrotalcite phase. No data have been found in the literature on this subject; but because it was prepared by the hydrotalcite synthesis procedure, it was named “Pb, Al-hydrotalcite

material". In the present work, the probability of the formation of some amount of lead-containing hydrotalcite was not rejected a priori, but investigated by X-ray diffraction, TEM, SEM, and other techniques and statistically verified.

2.2. X-ray diffraction

The X-ray diffraction (XRD) analysis was performed using a General Electric XRD-5 Diffractometer, at $\text{CuK}\alpha$ irradiation, and a nickel filter, in the range $2\theta = 5^\circ\text{--}90^\circ$, 2θ rate 4°min^{-1} . The position of the maximum of each reflection in the X-ray diffraction pattern was detected automatically. Phase analysis was performed using the procedure described elsewhere [5–9]. The existence of each phase was examined by calculation of its unit cell parameters. The precise determination of the unit cell parameters is based on calculation of their values using characteristic d -spacings and approximation of the Nelson–Raily function $a = f(\theta)$ to $\theta = 90^\circ$ [10]. The reliability of the Nelson–Raily function is estimated by correlation analysis. The correlation coefficient, r_{yx} , for each unit cell parameter has been calculated [11]. If $r_{yx} = 1$, a linear relationship exists between $f(\theta)$ and the unit cell parameter (a_0 or c_0). A value of $r_{yx} \neq 1$ indicates statistical control of the significance of the functional relationship between $f(\theta)$ and a_0 (or c_0). In this case the correlation coefficient, r_{yx} , calculated from the experimental data should be compared with its corresponding critical value, $r_{yx}(\alpha, f)$; where α is the level of significance, and $f = N - 2$ (N being the number of the parallel observations) is the number of degrees of freedom. In our investigations, the critical value of the correlation coefficient is $r_{yx}(\alpha, f) = 0.754$ [11]. If $r_{yx} > r_{yx}(\alpha, f)$, the relationship between $f(\theta)$ and the unit cell parameter is statistically significant, reliable, and can be described by polynomials of the first or second order [11].

2.3. Scanning electron microscopy

The scanning electron micrographs were recorded using a T-200 SEM at electron microscopic magnifications of 100–10 000, and photographic magnifications 1×2 .

2.4. Transmission electron microscopy

TEM investigations were performed with a JEM-100 at $U = 80 \text{ kV}$. A water suspension of finely dispersed powder of the Pb, Al-hydrotalcite material was prepared and investigated under "white-field" conditions at electron microscopic magnifications of 20 000, 30 000, 35 000, 50 000, 60 000, 70 000, 75 000, 80 000 and 100 000. The largest populations of particles observed were detected photographically and developed at photographic magnification 4×1 .

2.5. Infrared (IR) spectroscopy

IR spectra were recorded using a Bruker "IFS 113 V" Fourier Transform Infrared Spectrometer in the range

$4000\text{--}400 \text{ cm}^{-1}$ (standard error $\pm 1 \text{ cm}^{-1}$) in tablets, in air. About $0.0004 \pm 0.00005 \text{ g}$ of the sample were mixed in 0.200 g KBr (p.a.) and used to form tablets with surface of 1 cm^2 . Characteristic vibration data of the hydrotalcite structure [12, 13] and of other components of the hydrotalcite material [9] have been used in the interpretation. The characteristic vibrations of the structural OH groups, as well as of the water molecules and water complexes of the metal ions in the zeolite lattice were seen in the region $4000\text{--}3000 \text{ cm}^{-1}$. Their interpretation is possible only using the information from the diffuse-reflectance spectra.

2.6. Diffuse-reflectance spectroscopy (DRS)

The diffuse-reflectance spectra were recorded using a Beckman apparatus in the range $50\,000\text{--}4000 \text{ cm}^{-1}$. The pure material was dried at 120°C in a furnace for 24 h and cooled to room temperature in a desiccator, in the presence of blau-gel. Then tablets containing about 0.2 g of the material were prepared. As standard, tablets of BaSO_4 (p.a.) were used. The fundamental stretching vibrations, ν_{0-1} , and their first ν_{0-2} overtones, were calculated using the method described elsewhere [14–16].

3. Results

3.1. XRD

In the X-ray diffraction pattern of the uncalcined Pb, Al-hydrotalcite material (Fig. 1), reflections characteristic of dandassite $\text{PbAl}_2(\text{CO}_3)_4 \cdot 2\text{H}_2\text{O}$, hydrocerrussite $\text{Pb}_3(\text{CO}_3)_2(\text{OH})_2$ and $\text{Al}(\text{OH})_3$ gibbsite and bayerite, are seen. Weak and sharp reflections of hydrotalcite are also observed ($d_{003} = 0.682 \text{ nm}$, $d_{006} = 0.3787 \text{ nm}$, $d_{101} = 0.2616 \text{ nm}$, $d_{104} = 0.2443 \text{ nm}$, $d_{015} = 0.2311 \text{ nm}$, $d_{107} = 0.2072 \text{ nm}$, $d_{1010} = 0.1732 \text{ nm}$, $d_{113} = 0.1507 \text{ nm}$, $d_{116} = 0.1425 \text{ nm}$). The approximation of the linear Nelson–Raily function to $\theta = 90^\circ$ gives a slope for $a_0 = 0.30971 \text{ nm}$ ($r_{yx} = 0.9760$; standard deviation $\sigma = 0.00258$) and for $c_0 = 2.19196 \text{ nm}$ ($r_{yx} = 0.9258$; standard deviation $\sigma = 0.03795$). The corresponding critical value of r_{yx} (0.05, 5) is 0.754 [11]. The correlation coefficients $r_{yx} > 0.754$ for both unit cell parameters confirm the statistical significance and reliability of the linear

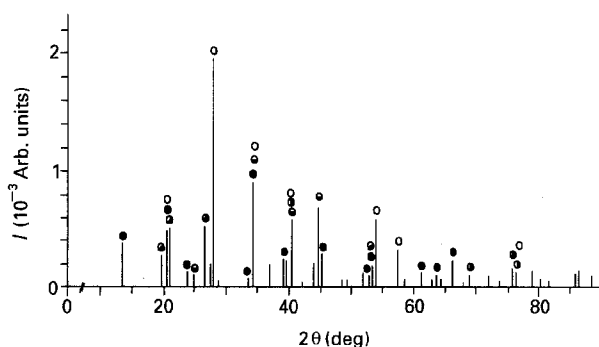


Figure 1 X-ray diffraction pattern of the uncalcined Pb, Al-hydrotalcite material: (●) hydrotalcite, (◐) $\text{Pb}_3(\text{CO}_3)_2(\text{OH})_2$, (◑) $\text{PbAl}_2(\text{CO}_3)_4 \cdot 2\text{H}_2\text{O}$, (○) $\text{Al}(\text{OH})_3$ bayerite, (◒) $\text{Al}(\text{OH})_3$ gibbsite.

TABLE I Millers' indexes (hkl), experimentally detected d -spacings d_{hkl} , experimental ("exp") and theoretical ("calc") values of the unit cell parameters, a_0 and c_0 , of the hydrotalcite structure

(hkl)	d_{hkl} (nm)	$a_{0\text{exp}}$ (nm)	$a_{0\text{calc}}$ (nm)	$c_{0\text{exp}}$ (nm)	$c_{0\text{calc}}$ (nm)
003	0.6820	—	—	—	—
006	0.3787	—	—	—	—
101	0.2616	0.30900 ± 0.0006	0.30732	2.36364 ± 1.5444	2.34620
104	0.2443	0.30675 ± 0.0004	0.30751	2.33918 ± 2.1207	2.33370
015	0.2311	0.31225 ± 0.0075	0.30766	2.32422 ± 1.1744	2.32422
107	0.2072	0.30873 ± 0.0050	0.30793	2.29116 ± 1.4358	2.30665
1010	0.1732	0.30874 ± 0.0051	0.30830	2.27030 ± 1.2196	2.28311
113	0.1507	0.30896 ± 0.0052	0.30862	2.26842 ± 1.8650	2.26242
116	0.1425	0.30674 ± 0.0059	0.30872	2.26842 ± 1.7650	2.25581

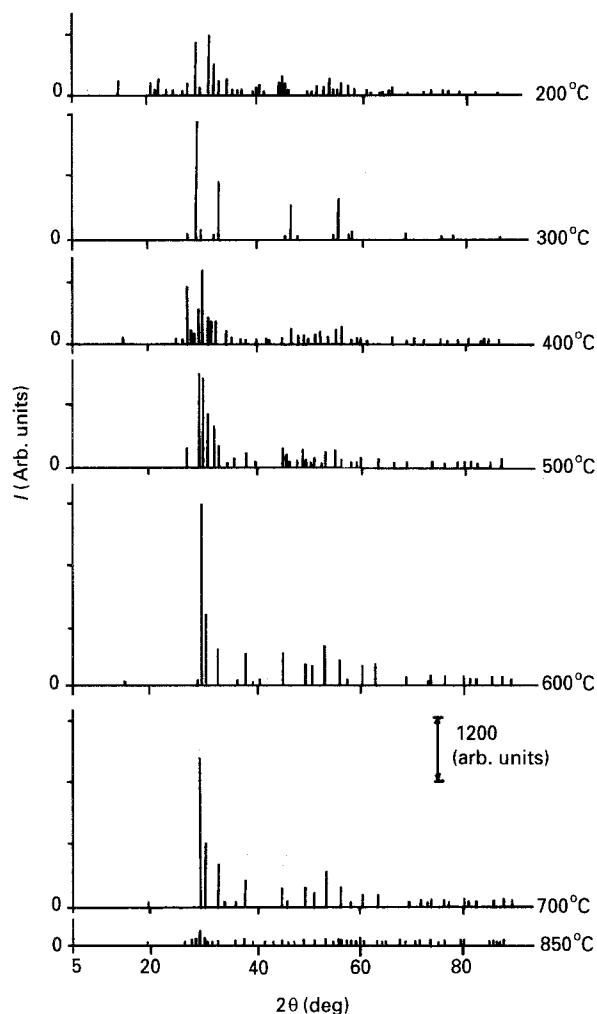


Figure 2 X-ray diffraction patterns of the Pb, Al-hydrotalcite material after calcination at 200–850 °C.

relationship between $f(\theta)$ and unit cell parameters. The d -spacings registered d_{hkl} , their corresponding Miller's indexes (hkl), and the experimental and theoretical values of the unit cell parameters, a_0 and c_0 , are given in Table I. The low intensities of the hydrotalcite reflections can be related to the small amount of this compound, in agreement with the small probability to form hydrotalcite structure with the participation of the Pb^{2+} ion with a large radius.

Several solid–solid phase transitions are registered after calcination of the material at 200–850 °C (Fig. 2). After calcination at 200 °C the characteristic reflections of the two polymorphous forms of $\text{Al}(\text{OH})_3$ relatively decrease (Fig. 3, curves 1, 2) and character-

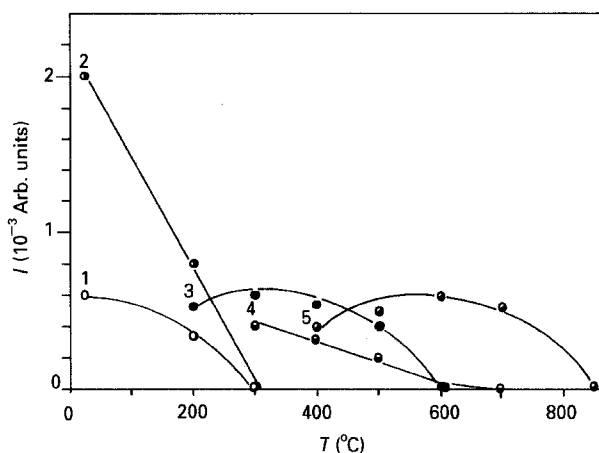


Figure 3 Relative changes in the intensities of the characteristic reflections of the (1, ○), gibbsite, (2, ●) bayerite, (3, ●) boehmite, (4, ●) $\alpha\text{-Al}_2\text{O}_3$, and (5, ●) $\gamma\text{-Al}_2\text{O}_3$ in the X-ray diffraction pattern of the Pb, Al-hydrotalcite material after calcination at 200–850 °C.

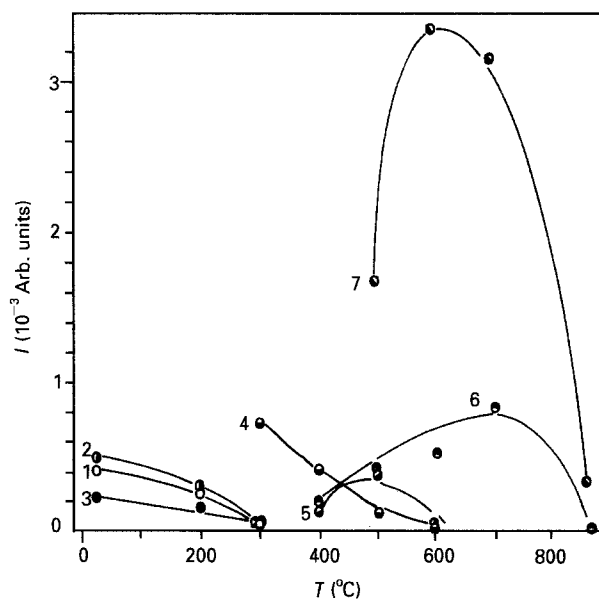


Figure 4 Relative changes in the intensities of the characteristic reflections of (1, ○) hydrotalcite, (2, ●) hydrocerrussite, (3, ●) dandassite, (4, ●) Pb_3O_4 , (5, ●) PbO_2 , (6, ●) $\alpha\text{-PbO}$, and (7, ●) $\beta\text{-PbO}$ in the X-ray diffraction patterns of the Pb, Al-hydrotalcite material after calcination at temperatures between 200 and 850 °C.

istic reflections of boehmite appear (Fig. 3, curve 3). The reflections of all lead-containing components of the material also relatively decrease (Fig. 4, curves 1–3). After calcination at 300 °C the character-

reflections of the lead-containing hydroxycarbonates (Fig. 4, curves 1–3), of the gibbsite (Fig. 3, curve 1) and bayerite (Fig. 3, curve 2), disappear. The reflections of the boehmite relatively increase (Fig. 3, curve 3) and characteristic reflections of α -Al₂O₃ (Fig. 3, curve 4) and Pb₃O₄ (Fig. 4, curve 4) appear. At 400 °C, the intensities of the boehmite (Fig. 3, curve 3) and Pb₃O₄ (Fig. 4, curve 4) decrease. Three new phases are formed at this temperature: γ -Al₂O₃ (Fig. 3, curve 5), α -PbO (Fig. 4, curve 6) and PbO₂ (Fig. 4, curve 5). After calcination at 500 °C, sharp and intensive characteristic reflections of β -PbO can be seen in the XRD pattern of the Pb, Al-hydrotalcite material (Fig. 4, curve 7). The intensities of the γ -Al₂O₃ (Fig. 3, curve 5), α -PbO (Fig. 4, curve 5) and PbO₂ (Fig. 4, curve 6) significantly increase, while those of Pb₃O₄ (Fig. 4, curve 4), boehmite (Fig. 3, curve 3) and α -Al₂O₃ (Fig. 3, curve 4) significantly decrease. At 600 °C the boehmite, α -Al₂O₃, Pb₃O₄ and PbO₂ characteristic reflections disappear. The characteristic reflections of the γ -Al₂O₃ (Fig. 3, curve 5) and of the β -PbO (Fig. 4, curve 7) became most intensive. At 700 °C the characteristic reflections of the γ -Al₂O₃ decrease insignificantly (Fig. 3, curve 5), while the intensities of the β -PbO reflections significantly decrease (Fig. 4, curve 7). After calcination at 850 °C, all characteristic reflections in the X-ray diffraction pattern of the sample

drastically decrease and the background level significantly increases. This can result even with an increase in dispersity, and even of amorphization.

3.2. SEM

Scanning electron micrographs of the uncalcined material show gibbsite, bayerite, crystal concretions of lead- and lead, aluminium hydroxycarbonates (Fig. 5a) and hydrotalcite (Fig. 5b). No amorphous phases are observed even at high electron microscopic magnifications. Unlike the Mg, Al-hydrotalcite material (Fig. 6a), in the scanning electron micrographs of the 700 °C calcined Pb, Al-hydrotalcite material (Fig. 6b), no amorphization is observed. The small crystallites adhere together forming systems of small channels of diameter 2 nm, and large channels of diameter 5 nm (Fig. 7a). On the walls of the large channels crystals of γ -Al₂O₃ are observed. The walls of the small channels are formed by β -PbO and α -Al₂O₃ crystals (Fig. 7b, c). It appears that decreasing intensity of the α -Al₂O₃ characteristic reflections in the X-ray diffraction pattern of the 700 °C calcined material is a result of increasing dispersity. On the basis of the SEM investigations it can be proposed that the decrease in the intensities of the characteristic reflections of all phases in the X-ray diffraction pattern of

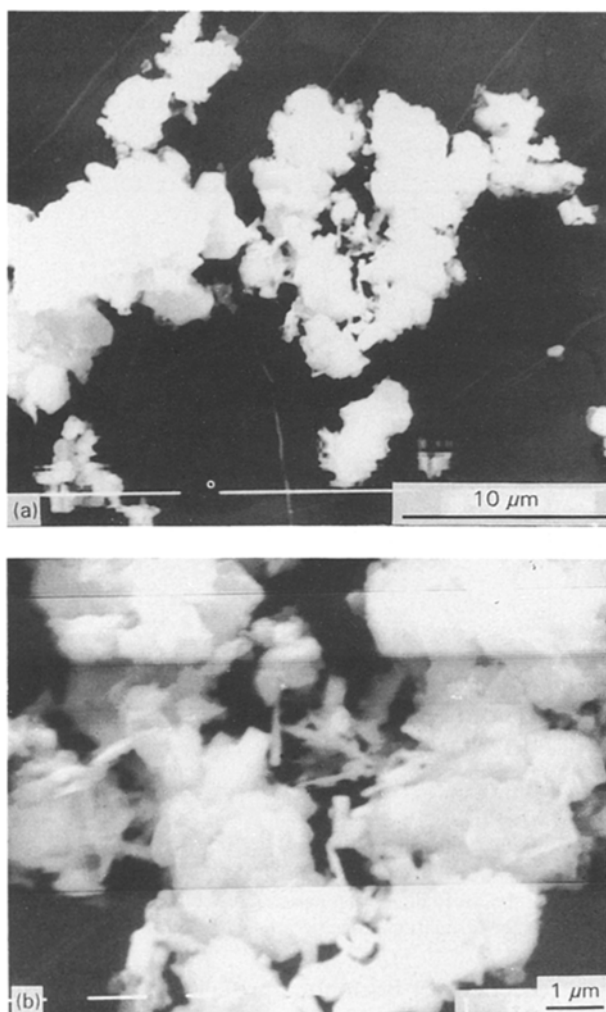


Figure 5 Scanning electron micrographs of the uncalcined Pb, Al-hydrotalcite material (a) $\times 3500$ and (b) $\times 10000$.

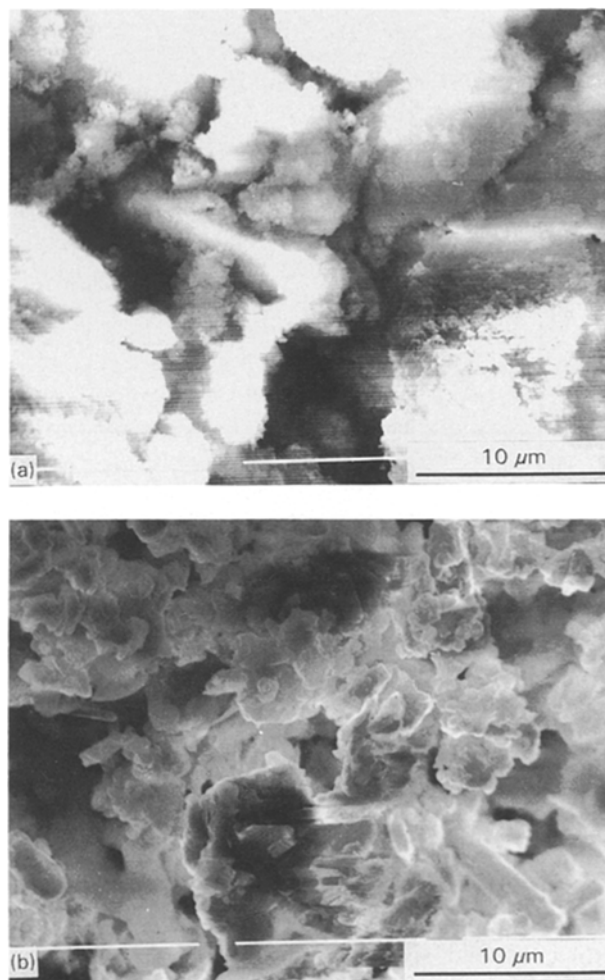


Figure 6 Scanning electron micrographs of the 700 °C calcined (a) Mg, Al-hydrotalcite material and (b) Pb, Al-hydrotalcite material; $\times 10000$.

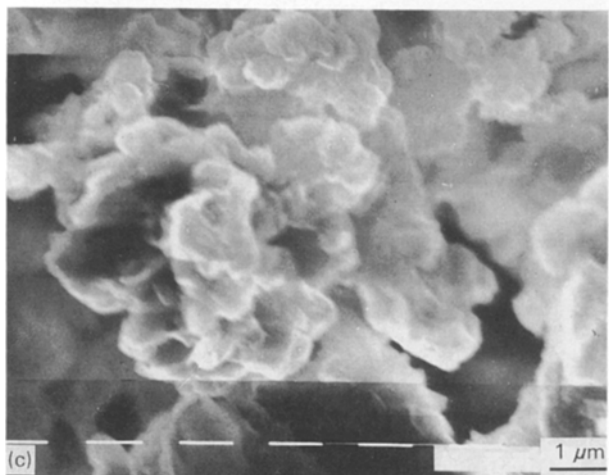
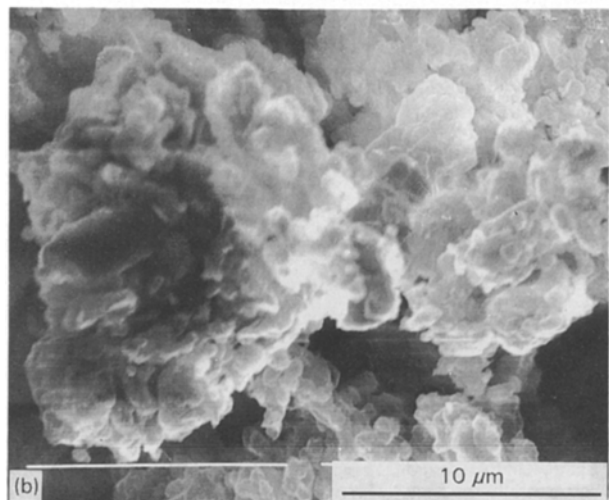
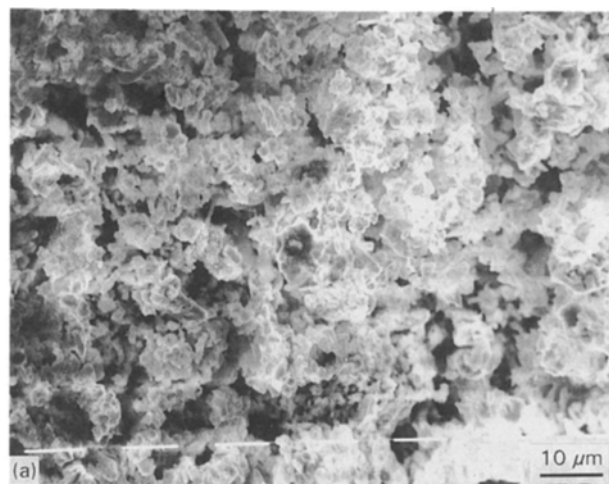


Figure 7 Scanning electron micrographs of 700°C calcined Pb, Al-hydroxalcite material at (a) $\times 1000$, (b) $\times 5000$, and (c) $\times 10000$.

the 850°C calcined material are the result of increased dispersity of the crystallites.

3.3. TEM

The transmission electron micrograph of the uncalcined Pb, Al-hydroxalcite material (Fig. 8a) shows a Japanese twine of hydroxalcite crystals, adhered to a hexagonal bipyramide II $\{1222\}$; on the bipyramidal face of one of them a parallel triple concretion of hydroxalcite is fixed; a crystal concretion of gibbsite is crystallized along the pinacoidal face of the

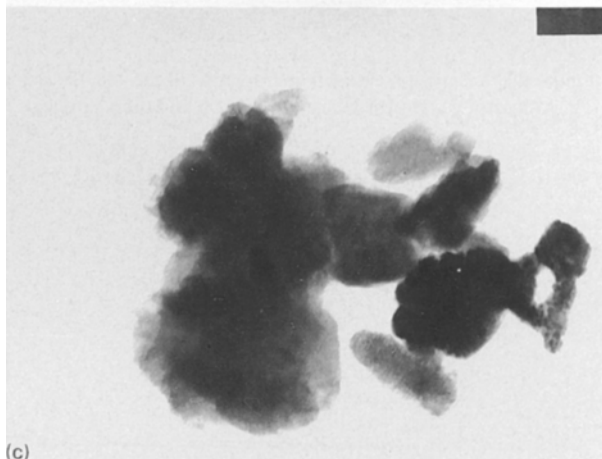
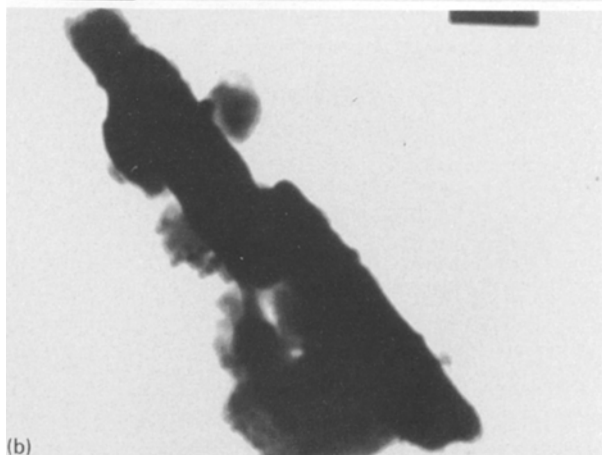
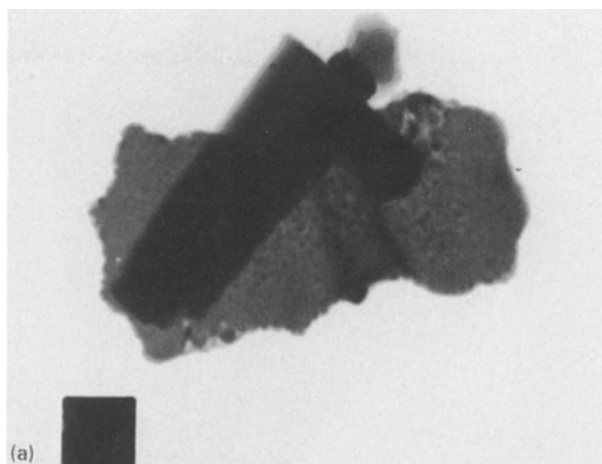


Figure 8 Transmission electron micrographs of the uncalcined Pb, Al-hydroxalcite material at (a) $\times 50000$, (b) $\times 70000$ and (c) $\times 60000$.

hydroxalcite. Fig. 8b shows a parallel crystal concretion of hydroxalcite in close vicinity to hydroxalcite, small crystals of dandassite and $\text{Al}(\text{OH})_3$. Gibbsite, bayerite, crystal concretions of hydroxalcite and dandassite in an $\text{Al}(\text{OH})_3$ matrix are also seen (Fig. 8c). The transmission electron micrographs of the 700°C calcined material show $\beta\text{-PbO}$ crystal concretions, forming a well-discernible channel system (Fig. 9a), concretions of $\alpha\text{-}$ and $\gamma\text{-Al}_2\text{O}_3$ with a well-developed surface (Fig. 9b).

3.4. IR

Fig. 10 shows the i.r. spectra of the uncalcined Pb, Al-hydroxalcite (spectrum 1) and Mg, Al-hydroxalcite

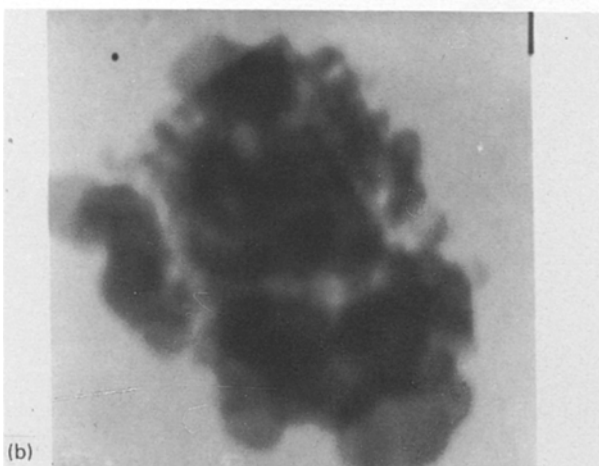
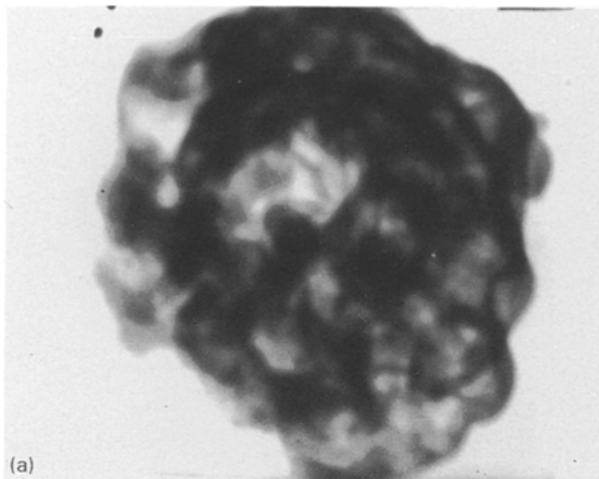


Figure 9 Transmission electron micrographs of the 700 °C calcined Pb, Al-hydroxalcite material at (a) $\times 50\,000$ and (b) $\times 80\,000$.

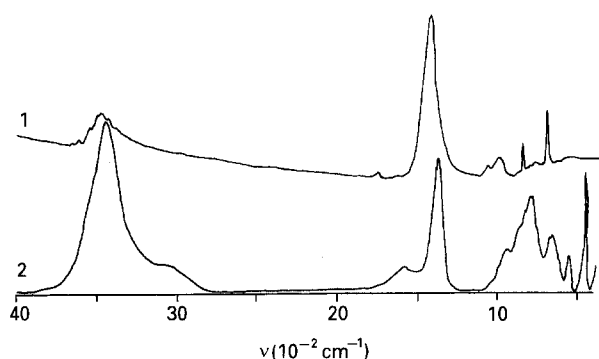


Figure 10 Infrared spectra of the uncalcined (1) Pb, Al-hydroxalcite material and (2) Mg, Al-hydroxalcite material.

(spectrum 2). The weak characteristic bands, corresponding to a hydroxalcite structure observed in the i.r. spectrum of the lead-containing material are shifted to the higher frequency side. The shift can be described by the different ionic radii of Mg^{2+} and Pb^{2+} cations [8].

3.5. DRS

The diffuse-reflectance spectrum of the uncalcined Pb, Al-hydroxalcite material is presented in Fig. 11. The band at about 2260–2240 nm ($4424\text{--}4464\text{ cm}^{-1}$) in the spectrum is attributed to the $(\nu + \delta)$ vibration of the $\text{CO}_3^{2-}\text{--H}_2\text{O}$ bond [14, 15] in the complexes $[\text{CO}_3(\text{H}_2\text{O})_4]^{2-}$ fixed in the interlayer distance of the

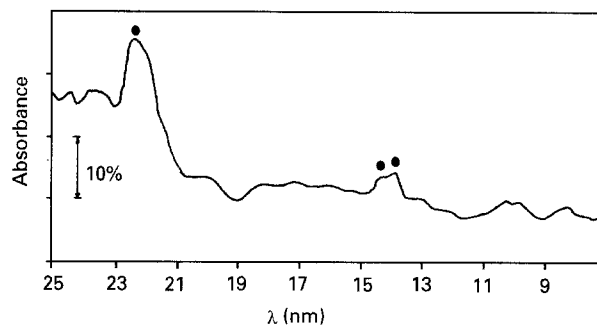


Figure 11 Diffuse-reflectance spectrum of the uncalcined Pb, Al-hydroxalcite material.

hydroxalcite structure [12, 13]. The $\nu_{0\rightarrow 1}$ vibration of this bond is at about 3050 cm^{-1} [16]. The shift of the band at 4424 cm^{-1} in the spectrum of the hydroxalcite can be the result of $\text{CO}_3^{2-}\text{--OH}$ bond length alteration as an effect of the interlayer distance alteration in the hydroxalcite structure. The position of this band in the DRS spectrum of the Pb, Al-hydroxalcite material is at 2240 nm, possibly as the effect of the relative decrease in the interlayer distance. In the XRD-pattern of the uncalcined Mg, Al-hydroxalcite material, an interlayer distance of 0.76 nm is detected, while in the XRD pattern of the lead-containing sample, the interlayer distance is 0.67 nm.

The band at 1450 nm (6896 cm^{-1}) corresponds to $\nu_{0\rightarrow 2}$ O–H bond vibration of $\nu_{0\rightarrow 1}$ 3525 cm^{-1} in the fragments $[\text{Me}^{2+}\text{--}(\text{O--H})\text{--Al}^{3+}]$ of the hydroxalcite structure [12–16]. The shift of the band at 1450 nm could be the result of O–H bond length alteration in the fragments $[\text{Me}^{2+}\text{--}(\text{O--H})\text{--Al}^{3+}]$ affected by the Me^{2+} ionic radius and electronegativity. This band is shifted to 1430 nm in the spectrum of the lead-containing hydroxalcite. It corresponds to the characteristic O–H bond vibration at 3575 cm^{-1} (seen as a weak band in the i.r. spectrum) in the fragment $[\text{Pb}^{2+}\text{--}(\text{OH})\text{--Al}^{3+}]$ of the Pb, Al-hydroxalcite.

The band at 1380 nm (7246 cm^{-1}) has been attributed [14–16] to $\nu_{0\rightarrow 2}$ of the characteristic stretching O–H bond vibration (3700 cm^{-1}) of the water molecules, coordinatively bonded to polyvalent cations.

4. Discussion

The results in the present work show that after hydroxalcite synthesize using lead and aluminium precursors, two polymorphous modifications of $\text{Al}(\text{OH})_3$ (gibbsite and bayerite) and three lead-containing hydroxycarbonates (dandassite, hydrocerrussite and hydroxalcite) are formed. The small amount of lead-containing hydroxalcite crystals is probably stabilized by the existence of gibbsite and hydrocerrussite in the system. During calcination, the aluminium hydroxides are transformed mainly to $\gamma\text{-Al}_2\text{O}_3$. The dehydration and decarboxylation of the lead-containing components of the material are accomplished at $300\text{ }^\circ\text{C}$, forming a well-developed large-channel system in the calcined sample. The complicated phase transitions of the boehmite and lead oxides above $400\text{ }^\circ\text{C}$, accompanied by dehydration and loss of oxygen, evidently

result from the formation of the small-channel system and crystallization of small crystals of α - Al_2O_3 . The calcination of the Pb, Al-hydroxalcite material at 600–750 °C leads to the formation of β -PbO and γ - Al_2O_3 . On the basis of the present and previously reported [2, 3] investigations it can be assumed that calcination at 600–750 °C is the best catalytic pre-treatment for the Mg, Pb, Al-hydroxalcite material, at which mainly the catalytically active methane dimerization β -PbO phase is formed in highly dispersed MgO– Al_2O_3 matrix of high basicity.

5. Conclusions

1. By synthesis of hydroxalcite using Reichle's procedure and lead and aluminum nitrates, material is obtained containing gibbsite, bayerite, hydroxalcite, dandassite and a small amount of Pb, Al-hydroxalcite. The hydroxalcite crystals are stabilized in the presence of gibbsite and hydroxalcite.

2. The complete dehydration and decarboxylation of the lead-containing components of the Pb, Al-hydroxalcite material at 300 °C is accompanied by the formation of a well-developed large-channel system in the sample. The small-channel system in the material is formed above 400 °C, during the thermal degradation of boehmite which is accompanied by the release of water ($2\text{AlOOH} \rightarrow \text{Al}_2\text{O}_3 + \text{H}_2\text{O}$), and the phase transitions of the lead oxides which is accompanied by loss of oxygen ($2\text{PbO}_2 \rightarrow \text{PbO} + \text{O}_2$).

3. The calcination of the Pb, Al-hydroxalcite material at 600–750 °C leads to the formation mainly of β -PbO and γ - Al_2O_3 .

4. The calcination of the Mg, Pb, Al-hydroxalcite material at temperatures 600–750 °C is the best catalytic pre-treatment, leading formation of the catalytically active methane dimerization β -PbO phase in a highly dispersed MgO– Al_2O_3 matrix of high basicity.

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